



# Promoting industrial-level CO<sub>2</sub> electroreduction kinetics via accelerating proton feeding on a metal-free aerogel electrocatalyst

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## ABSTRACT

Developing cost-effective carbon based electrocatalysts with high activity and selectivity for CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) at industrial current density still remains challenges. Herein, we present a metal-free 3D cross-linked carbon aerogel catalyst containing homogeneously dispersed N and P sites (P@NCA) synthesized via freeze drying and subsequent carbonization. Benefit from the synergistic effect of N and P atoms, P@NCA aerogel delivers an excellent CO<sub>2</sub>RR activity featured by a wide potential window from −0.3 V to −0.9 V with CO Faradaic efficiency (FE) of above 90 %. Importantly, a high CO FE of 85 % at an industrial current density of 100 mA cm<sup>−2</sup> is realized. Experimental explorations and in-situ spectroelectrochemistry analyses reveal that pyrrolic-N species acts as the real active site to execute CO<sub>2</sub>RR, while the P species facilitates the protonation process of \*CO<sub>2</sub> to \*COOH intermediate by accelerating the proton feeding from water dissociation. Theoretical calculations demonstrate a suppressed \*H adsorption and a facilitated \*CO desorption step on pyrrolic-N site with P doping, boosting the whole CO<sub>2</sub>RR process. Further, a rechargeable Zn-CO<sub>2</sub> battery constructed with P@NCA aerogel as cathode delivers a maximal power density of 0.8 mW cm<sup>−2</sup> and a superior stability.

## 1. Introduction

Electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) driven by renewable energy is a promising technology to reduce CO<sub>2</sub> concentration in atmosphere [1,2]. Compared with industrially energy-intensive thermocatalytic reactions, electrocatalytic CO<sub>2</sub>RR process can convert CO<sub>2</sub> molecules into high value-added chemicals (CO, ethylene, and formic acid, etc.) under mild conditions [3,4]. In recent years, noble metal and its alloy catalysts, such as Au, Ag, Pd, PdCu, and AgCu, etc., have made good progress in the field of CO<sub>2</sub>RR [5]. Especially, carbon-supported metal single atom catalysts (M-N-C, M = Ni, Fe, Co, Cu, and Mn, etc.) presented superior catalytic activity for CO<sub>2</sub>RR at high current density in the designed electrolyzers [6,7]. However, these mentioned metal-based

catalysts still face several inevitable disadvantages, such as low abundance of earth elements, high cost, and complicated synthesis steps, etc., which seriously hinder the practical industrial utilization of CO<sub>2</sub>RR [8,9].

In contrast, metal-free carbon based materials, especially the electrocatalysts featured with unique hierarchical interconnected structures [10] that possess the advantages of low cost, resistance to acids and bases, porous structure with various pores, and environmental friendliness, have a good prospect for the practical application of CO<sub>2</sub>RR [11–13]. However, previously reported non-metallic carbon based CO<sub>2</sub>RR catalysts, in particular, the widely reported N-doped carbon electrocatalysts, face inevitable problems, such as monotonous active site, slow CO<sub>2</sub> reaction kinetics, and strong competitive hydrogen

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evolution reactions (HER), thus resulting in low CO<sub>2</sub>RR activity [9,14]. Despite the simple synthesis method, N-doped carbon materials still display poor catalytic performance in CO<sub>2</sub>RR, which is difficult to achieve industrial current density and wide potential window. Moreover, the carbon catalyst doped only with N element cannot achieve the desired catalytic performance due to the strong \*H adsorption on N sites that often causes competitive HER [15,16]. The introduction of additional heteroatoms (P, S, F, and Se, etc.) can efficiently change the electronic structure of N-doped carbon catalysts. Under the synergistic effect of di-heteroatoms (N/S, N/P, N/Se, N/F, etc.), the adsorption strength of key intermediates (\*COOH or \*CO) of CO<sub>2</sub>RR on active site will be enhanced or weakened, thus achieving catalytic performance enhancement [8,17], which has considered as an effective method to boost CO<sub>2</sub>RR activity and suppress the HER [11,18]. Compared with other heteroatoms (S, Se, and F), foreign P atom has the largest atomic radius and lowest electronegativity ( $P < Se < S < F$ ), thus the doping of P atom would possess the highest possibilities of adjusting the electronic structure of N atoms, thereby optimizing the adsorption free energy of key intermediates in CO<sub>2</sub>RR. Despite certain progress on CO<sub>2</sub>RR catalyzed by di-heteroatoms doped carbon catalysts, the catalytic current density is still low in most cases. Especially, achieving high CO Faradaic efficiency (FE) under industrial current density ( $\geq 100 \text{ mA cm}^{-2}$ ) and wide potential window are still challenging in CO<sub>2</sub>RR, meanwhile, understanding the reaction mechanism and identifying actual active site of di-heteroatoms in CO<sub>2</sub>RR are urgently needed [19,20].

Herein, we report a metal-free carbon aerogel CO<sub>2</sub>RR catalyst featuring diatomic N-P sites (P@NCA) which was prepared via a radical polymerization, followed by a high temperature carbonization treatment under ammonia atmosphere. Benefited by the unique 3D interconnected structure with homogeneous meso-pores distribution and highly exposed N, P sites, P@NCA aerogel displays high CO FE of above 90 % at a wide potential window from  $-0.3 \text{ V}$  to  $-0.9 \text{ V}$  via CO<sub>2</sub>RR. As-prepared P@NCA aerogel also achieves an industrial-level current density of  $100 \text{ mA cm}^{-2}$  with a CO FE of 85 %, outperforming a majority of other reported metal-free carbon based CO<sub>2</sub>RR electrocatalysts. A series of controlled experiments and structural characterizations demonstrate the pyrrolic-N species acted as actual active sites in CO<sub>2</sub>RR. In-situ attenuated total reflectance-Fourier transform infrared (ATR-FTIR) confirms under the regulation of surrounding P atoms, the water dissociation step and subsequent CO<sub>2</sub> protonation process (\*CO<sub>2</sub> → \*COOH) on pyrrolic-N site is significantly promoted. Theoretical calculations reveal that the P dopants induce the increase of charge density over the pyrrolic-N site, which enhances the CO desorption from active pyrrolic-N, thereby improving the CO<sub>2</sub>-CO selectivity. When assembled as cathode in rechargeable Zn-CO<sub>2</sub> battery, the newly-developed P@NCA aerogel display a high power density of  $0.8 \text{ mW cm}^{-2}$  and long-term stability of 20 h for CO production.

## 2. Experimental section

### 2.1. Synthesis of P@NCA aerogel

2.76 mL of phytic acid (50 wt% in water) and 6 mL of deionized water (DI water) were added into a bottle under vigorous stirring for 5 min. Then, 1.65 mL of aniline solution was added into above bottle and stirring for 10 min. Next, 0.855 g of ammonium persulfate (APS) was dissolved in 3 mL DI water. After cooling to 4 °C, the APS solutions were added into the bottle and keep stirring for 10 min. The fully reacted mixture was transferred into the refrigerator and kept at 4 °C for 12 h to obtain P@NCA hydrogels precursor. Then, the obtained precursor was centrifuged and washed three times with anhydrous ethanol and DI water, respectively, and then freeze-drying for 24 h under vacuum. Finally, the sample was put in a tube furnace and heated to 1000 °C with a rate of  $3 \text{ °C min}^{-1}$ , then kept for 1 h in 10 % NH<sub>3</sub>/N<sub>2</sub> atmosphere. After cooling to room temperature, the P@NCA aerogel was obtained.

### 2.2. Synthesis of NCA aerogel

The synthesis procedure for NCA aerogel is similar to that of P@NCA aerogel, except adding the phytic acid, the obtained sample was named as NCA aerogel.

### 2.3. Synthesis of P@CA aerogel

2.76 mL of phytic acid (50 wt% in water) and 6 mL of DI water were added into a bottle under vigorous stirring for 5 min. Next, 200 mg of carbon black with 20 mL of DI water was added into above bottle and keeps stirring for 2 h. Then, the formed mixture was placed in a centrifuge tube and freeze-dried for 24 h. Finally, the obtained sample underwent a same carbonization process with P@NCA, the sample of P@CA aerogel was obtained.

### 2.4. Electrochemical measurements

The performance of electrochemical CO<sub>2</sub> reduction over as-prepared catalysts was performed in flow cell and H-cell equipped with three-electrode system on the electrochemical work station (CHI 760E). The electrolyte in H-cell and flow cell was CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> and 1.0 M KHCO<sub>3</sub> solution, respectively. Both the anodic and cathodic electrolyte chambers in H-cell and flow cell were separated by the cation exchange membrane (Nafion 117, DuPont). In H-cell, the highly purified CO<sub>2</sub> gas (99.999 %) was injected into the cathodic electrolyte with a flow rate of  $20 \text{ mL min}^{-1}$ , while in flow cell, the CO<sub>2</sub> (99.999 %,  $20 \text{ mL min}^{-1}$ ) was injected into the chamber in the backside of working electrode. The reference electrodes used in H-cell and flow cell were Ag/AgCl electrode; and the counter electrode was Pt wire and Ni foam in H-cell and flow cell, respectively; the details of working electrode preparation can be found in [Supporting Information](#). The linear sweep voltammetry curve was performed under the range of 0 to  $-1.2 \text{ V}$  with the scan rate of  $5 \text{ mV s}^{-1}$ , the electrochemical impedance spectrum was tested under 10 mV AC voltage, along with the frequency range from 10 mHz to 10 KHz.

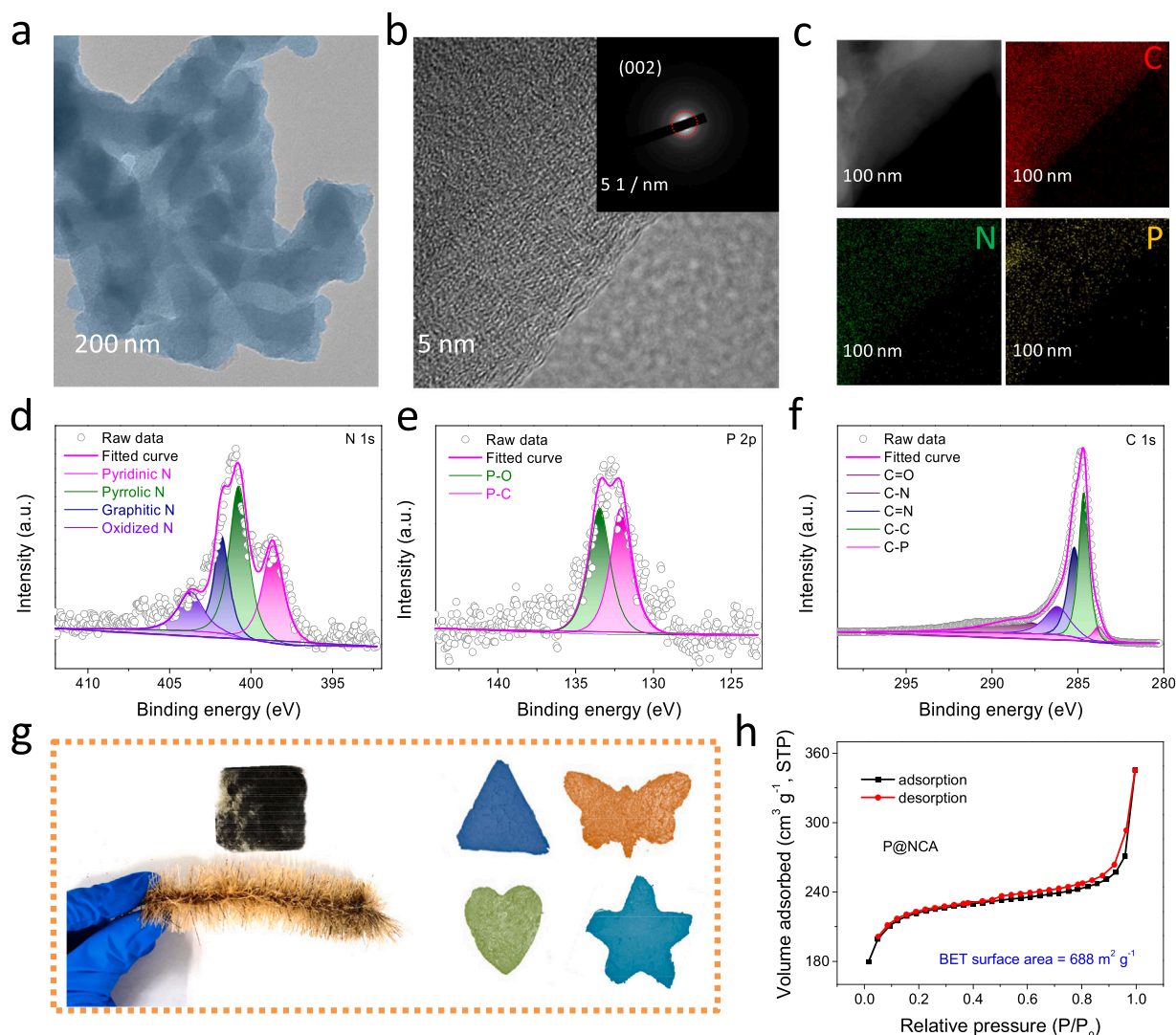
### 2.5. DFT analysis

The theoretical calculations were performed by using Vienna ab initio simulation package. A plane wave cutoff energy of 400 eV was used. The generalized gradient approximation proposed by Perdew, Burke, and Ernzerhof was utilized in the projector-augmented wave method. Additional details can be found in [Supporting Information](#).

## 3. Results and discussion

The 3D catalyst of metal free P@NCA aerogel was synthesized through a low temperature polymerization and high temperature carbonization method ([Fig. S1](#)), in which phytic acid and aniline monomer were selected as P and N sources, respectively [21]. When mixed two reactants, a protonating reaction of the N groups in aniline happened to generate phytic acid-aniline composite. After the oxidative initiator of ammonium persulfate added, corresponding polyaniline-phytic acid hydrogel was formed under the ionic-polymerization reaction ([Figs. S2–S4](#)) [22,23]. Followed by freeze drying and carbonization at 1000 °C in ammonia atmosphere, the P@NCA hydrogel was convert to produce the final product of P@NCA aerogel ([Figs. S5–S6](#)). During the carbonized process, the N and P atoms are gradually activated with the increase of temperature, and the evaporation of carbon leads to large surface area and highly exposed active sites, which can effectively enhance the performance of electrochemical CO<sub>2</sub>RR [24].

Transmission electron microscopy (TEM) image of P@NCA aerogel ([Fig. 1a](#)) showed a unique 3D cross-linked characteristic with particle size of about 100 nm [25]. Further high-resolution TEM (HRTEM) image

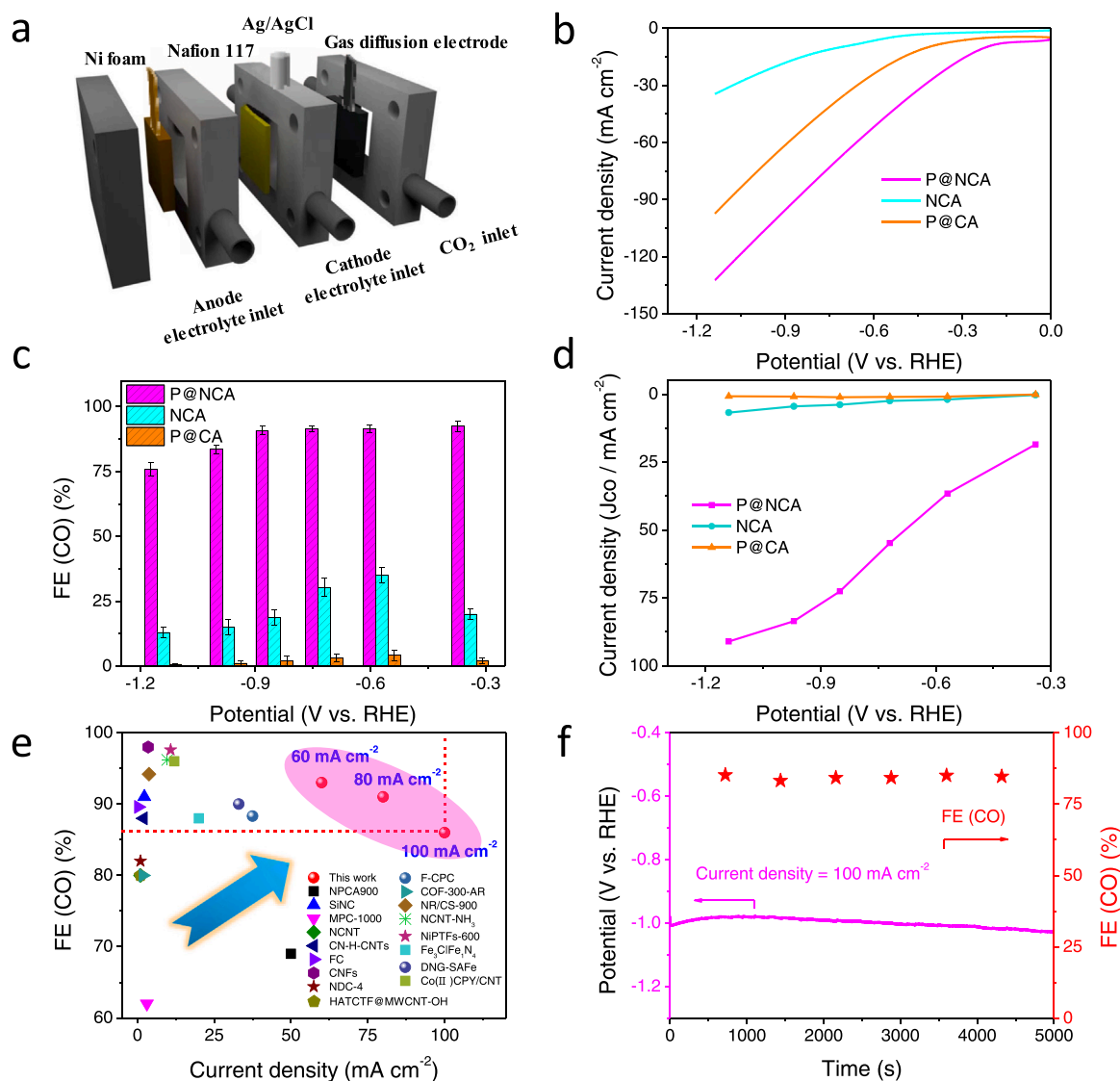


**Fig. 1.** (a–b) TEM and HRTEM images of P@NCA (inset: SAED pattern), (c) EDX elemental mappings of C, N, and P for P@NCA, (d–f) high resolution N 1s, P 2p, and C 1s XPS spectra of P@NCA, (g) digital photo of metal-free carbon aerogel prepared and its different shapes, (h)  $N_2$  adsorption-desorption isotherm curve of P@NCA.

of P@NCA aerogel displayed a lattice fringe distance of 0.34 nm in edge (Fig. 1b and Fig. S7), which is attributed to (002) plane of graphitized carbon [26,27]. The corresponding inset image of selected area electron diffraction (SAED) pattern exhibited a typical ring belong to the (002) plane of carbon, which is well supported by X-ray diffraction (XRD) results (Fig. S8) [28,29]. Energy-dispersive X-ray spectroscopy (EDX) showed a homogeneous distribution of C, N, and P elements in the P@NCA aerogel (Fig. 1c) [30]. X-ray photoelectron spectroscopy (XPS) spectra of P@NCA aerogel further confirmed the co-existence of C, N, O, and P elements (Figs. S9–S12). For the high-resolution N 1s XPS spectrum of P@NCA aerogel, four different types of N species were identified as pyridinic-N (398.6 eV), pyrrolic-N (400.7 eV), graphitic-N (401.8 eV), and oxidized-N (403.7 eV), respectively (Fig. 1d) [31]. The high-resolution P 2p XPS spectrum of P@NCA aerogel (Fig. 1e) showed two typical chemical bonds of P-C and P-O located at 132.1 eV and 133.4 eV, respectively [32]. The C-N bond (286.3 eV) and C-P bond (283.8 eV) were also observed in P@NCA aerogel as detected by high-resolution C 1s XPS spectrum (Fig. 1f). Both of them proved the existences of C-N and C-P bonds, suggesting the successful synthesis of P@NCA aerogel. Notably, the shapes of P@NCA aerogel can be molded into various models (Fig. 1g), for example, the cylindrical P@NCA model stranded on yellow flower, indicating the lightness of this aerogel, other prepared models of triangle, butterfly, heart, and pentagram,

indicating the processability of as-prepared aerogel [33]. Raman spectra of P@NCA aerogel showed an increased intensity ratio of D and G bands compared with other mono-doped samples, indicating the rich defects structure (Fig. S13) [34,35]. The Brunauer-Emmett-Teller (BET) surface area of P@NCA aerogel was determined to be  $688 \text{ m}^2 \text{ g}^{-1}$  with an average pore size of 6 nm (Fig. 1h and Fig. S14), demonstrating the mesoporous structure of P@NCA aerogel, in favor of  $\text{CO}_2$  transfer and adsorption [36].

The  $\text{CO}_2$ RR performances of P@NCA aerogel were tested in a flow-cell reactor to eliminate the mass transfer effects of  $\text{CO}_2$ , in which gas diffusion layer (GDL) loading the catalyst was used as working electrode (Fig. 2a and Fig. S15). For comparison, control samples of NCA (P-free) and P@CA (N-free) were synthesized (Figs. S16–S17). The gaseous and liquid products from  $\text{CO}_2$ RR were measured by on-line gas chromatography and  $^1\text{H}$  nuclear magnetic resonance spectroscopy (NMR), and the results showed no liquid products formed (Fig. S18). Linear sweep voltammetry (LSV) curve of P@NCA aerogel showed a much higher total current density than that of control P@CA and NCA in 1.0 M  $\text{KHCO}_3$  electrolyte (Fig. 2b). The CO and  $\text{H}_2$  FEs of three investigated catalysts were shown in Fig. 2c and Fig. S19. Among them, the P@NCA aerogel displayed excellent CO FEs of above 90 % from  $-0.3 \text{ V}$  to  $-0.9 \text{ V}$ , and the CO FE maintained at 85 % when a potential of  $-1.0 \text{ V}$  was conducted, much higher than the other control samples of P@CA and NCA,



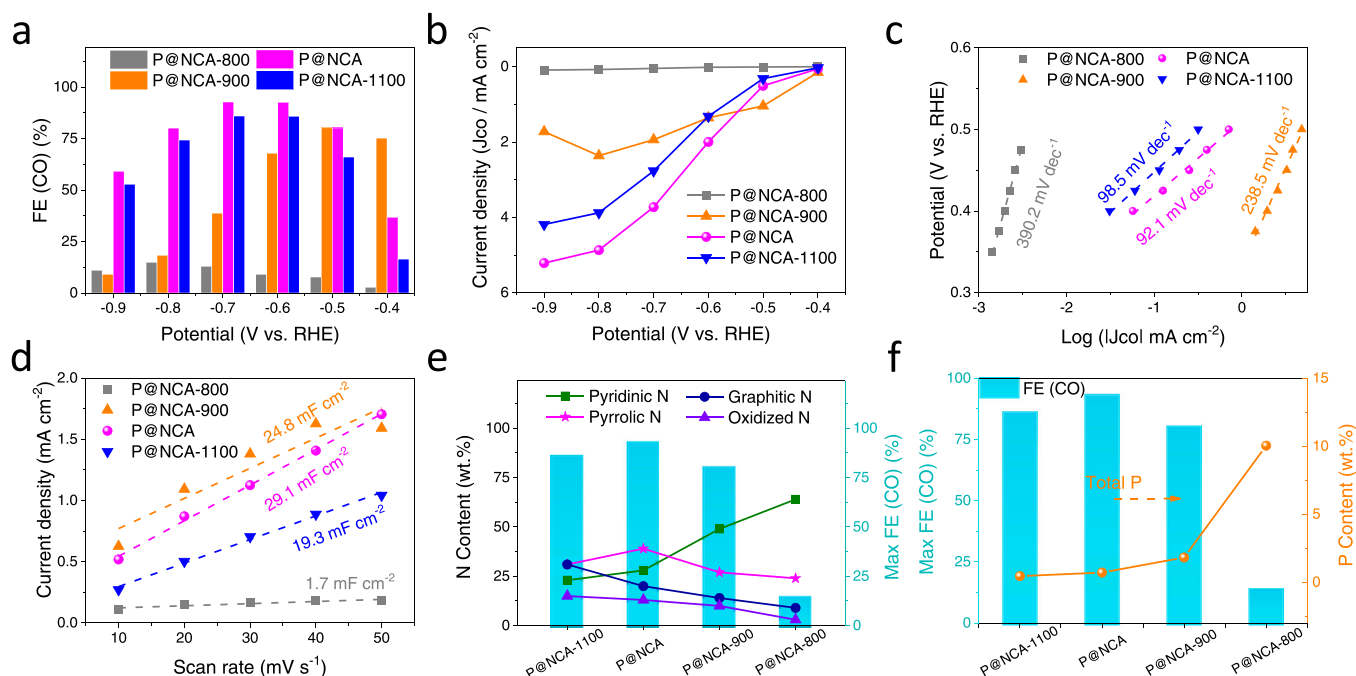
**Fig. 2.** (a) Assembly diagram of flow-cell reactor, (b) polarization curves of P@NCA, NCA, and P@CA, (c) CO FEs of P@NCA, NCA, and P@CA, (d)  $J_{CO}$  of P@NCA, NCA, and P@CA at different potentials, (e) CO<sub>2</sub>RR activity comparison of P@NCA with other reported metal-free or metal-N carbon catalysts, (f) stability test of P@NCA at 100 mA cm<sup>-2</sup>.

indicating its superior activity in CO<sub>2</sub>RR [37]. In addition, the partial current densities of CO ( $J_{CO}$ ) for the three catalysts were calculated. As shown in Fig. 2d, the P@NCA aerogel showed a much higher  $J_{CO}$  of 90 mA cm<sup>-2</sup> than the NCA (10 mA cm<sup>-2</sup>) and P@CA (1.2 mA cm<sup>-2</sup>) at an applied potential of -1.2 V, revealing the synergistic roles of N and P species in the P@NCA aerogel to boost the CO<sub>2</sub>RR activity enhancement. Compared with other previously reported state-of-the-art CO<sub>2</sub>RR catalysts from metal-free to metal-N carbon materials (Fig. 2e), the as-prepared P@NCA aerogel showed a top-of-the-range CO<sub>2</sub>RR performance in terms of CO FE and current density. Moreover, the catalytic stability of P@NCA aerogel at an industrial-level current density of 100 mA cm<sup>-2</sup> for CO<sub>2</sub>RR was tested, and no obvious decay in CO FE was observed during 5000 s continuous reaction (Fig. 2f and Fig. S20), suggesting a superb durability [38,39].

In order to explore the influence of pyrolysis temperature on the performance trend of CO<sub>2</sub>RR, control samples of P@NCA aerogels carbonized at different temperatures of 800, 900, and 1100 °C were prepared, denoted as P@NCA-800, P@NCA-900, and P@NCA-1100 [40]. And the corresponding electrochemical CO<sub>2</sub>RR tests were performed in H-type cell (Figs. S21–S26). Clearly, a peak-like trend of CO<sub>2</sub>RR activity from top to bottom with the change of temperature was

observed. Especially, the P@NCA aerogel (Fig. 3a) showed a much higher maximum CO FE of 94 % than the P@NCA-1100 (86 %), P@NCA-900 (76 %), and P@NCA-800 (13 %) at -0.7 V. The same peak-like trend of CO<sub>2</sub>RR activity can also be observed for  $J_{CO}$  in Fig. 3b, in which the P@NCA aerogel delivered a high  $J_{CO}$  of 5.2 mA cm<sup>-2</sup> at -0.9 V, which was much higher than the P@NCA-1100 (4.2 mA cm<sup>-2</sup>), P@NCA-900 (1.7 mA cm<sup>-2</sup>), and P@NCA-800 (0.1 mA cm<sup>-2</sup>) at the same potential, respectively. Tafel slope of P@NCA aerogel was calculated as 92.1 mV dec<sup>-1</sup> (Fig. 3c), which was lower than that of P@NCA-1100 (98.5 mV dec<sup>-1</sup>), P@NCA-900 (238.5 mV dec<sup>-1</sup>), and P@NCA-800 (390.2 mV dec<sup>-1</sup>), indicating a rapid CO<sub>2</sub> reaction kinetics on P@NCA aerogel. It is worth noting that the value of 92.1 mV dec<sup>-1</sup> is close to the theoretical value of CO<sub>2</sub> protonation step, indicating that the rate determining step of CO<sub>2</sub>RR over P@NCA aerogel is the process of CO<sub>2</sub> into \*COOH intermediate [41]. Moreover, among all of the investigated samples, the P@NCA aerogel delivered the lowest electron transport resistance (Fig. S27), revealing its good electron transfer capability in CO<sub>2</sub>RR process [42]. Further, the P@NCA aerogel showed a higher capacitance of double layer ( $C_{dl}$ ) value of 29.1 mF cm<sup>-2</sup> than the P@NCA-900 (24.8 mF cm<sup>-2</sup>), P@NCA-1100 (19.3 mF cm<sup>-2</sup>), and P@NCA-800 (1.7 mF cm<sup>-2</sup>), respectively, suggesting its largest





**Fig. 3.** (a) CO FEs and (b) corresponding  $j_{CO}$  of P@NCA-800, P@NCA-900, P@NCA, and P@NCA-1100 at different potentials in H-type cell, (c) Tafel slopes and (d) calculated  $C_{dl}$  values of P@NCA-800, P@NCA-900, P@NCA, and P@NCA-1100, (e) CO FEs and contents of fitted N species for P@NCA-800, P@NCA-900, P@NCA, and P@NCA-1100, (f) CO FEs and content of P species for P@NCA-800, P@NCA-900, P@NCA, and P@NCA-1100.

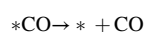
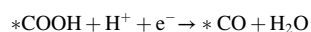
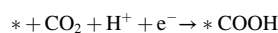
electrochemically active surface area in CO<sub>2</sub>RR (Fig. 3d and Fig. S28) [43].

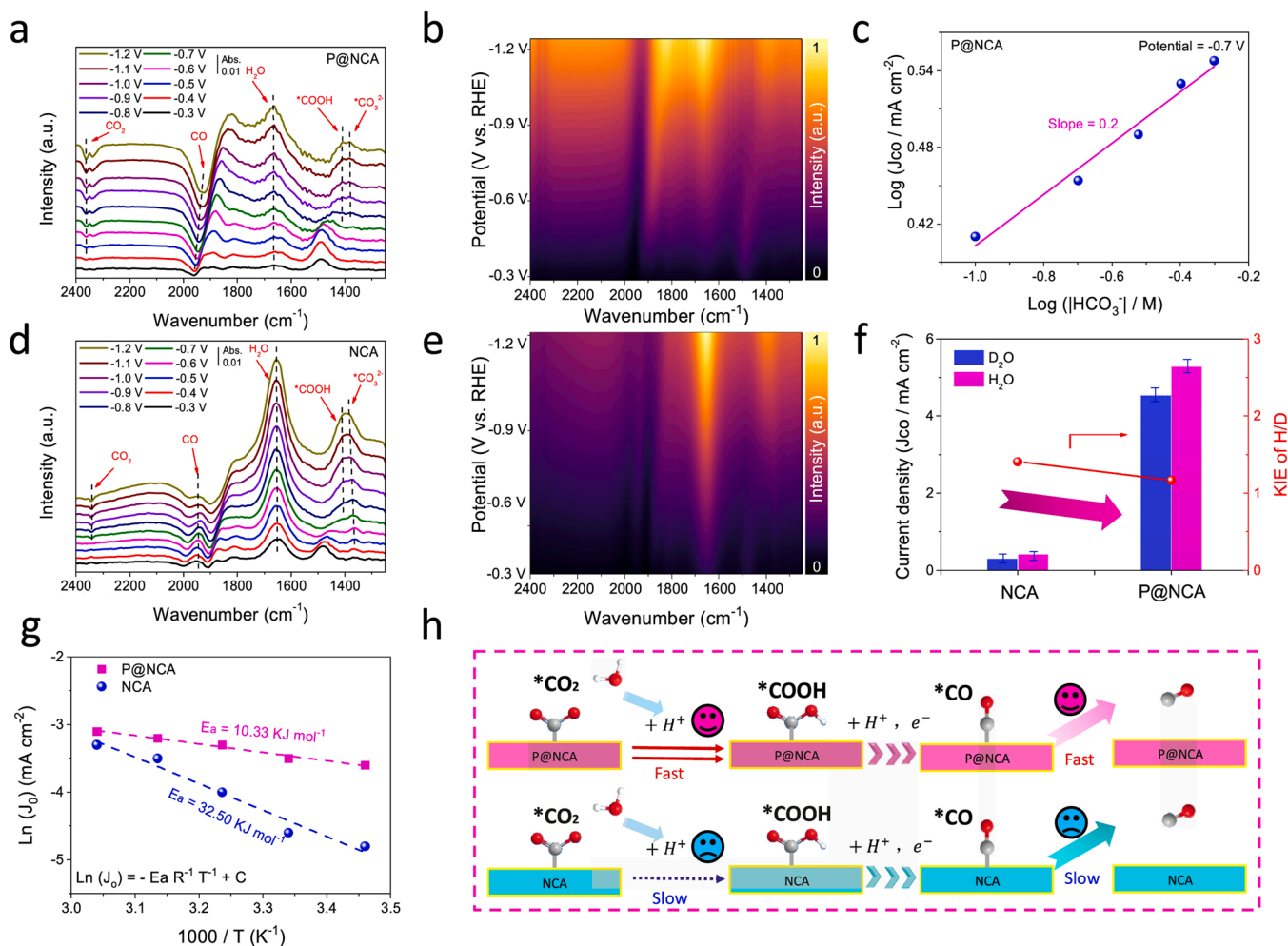
For further determining the actual active sites of P@NCA aerogel in CO<sub>2</sub>RR, the relationship between N, P contents and CO FEs was analyzed for P@NCA and P@NCA-800/900/1100 samples. In Fig. 3e and Fig. S29, four different types of N species were fitted by high-resolution N 1s XPS spectra, and we observed that only the content of pyrrolic-N was consistent well with the change trend of CO FE, indicating that the pyrrolic-N was involved in CO<sub>2</sub>RR as the real active sites. In contrast, the content of P species was uncorrelated with the changing trend of CO FE (Fig. 3f), revealing that the P species was not acted as main active sites that contributed to the adsorption and transformation reaction of CO<sub>2</sub> molecules. Based on this point, it is speculated that the P species possibly acted as a cocatalytic atom to accelerate the CO<sub>2</sub>RR kinetics via changing the electronic structure of pyrrolic-N [44].

In order to deeply discover the intrinsic promoting effect of doped P atoms on P@NCA aerogel for CO<sub>2</sub>RR, in-situ ATR-FTIR tests were performed from  $-0.3$  V to  $-1.2$  V (Fig. S30). As shown in Fig. 4a-b, three positive peaks located at 1395, 1420, and 1670  $\text{cm}^{-1}$  are assigned to the CO<sub>2</sub><sup>2-</sup>, \*COOH and H<sub>2</sub>O, respectively, indicating the generation of adsorbed \*COOH intermediates. At the same time, two negative peaks located at 2360 and 1960  $\text{cm}^{-1}$  are classified as CO<sub>2</sub> and \*CO respectively, demonstrating that both of them are gradually consumed during CO<sub>2</sub>RR process [45]. Among them, the \*COOH intermediate is continuously generated from  $-0.8$  V to  $-1.2$  V, while the CO<sub>2</sub> is gradually consumed, suggesting that the protonation process from CO<sub>2</sub> to \*COOH intermediate is rate-determining step. To further investigate the source of protons during protonation step of CO<sub>2</sub>, the  $j_{CO}$  of P@NCA aerogel were tested in different concentrations of KHCO<sub>3</sub> (Fig. S31). In Fig. 4c, the slope value was calculated to be 0.2 according to the linear relationship between the concentration of KHCO<sub>3</sub> electrolyte and  $j_{CO}$ , which revealed a zero-order dependence between CO<sub>2</sub>RR rate with KHCO<sub>3</sub> concentration. This result confirmed that the proton is mainly derived from water when the protonation step is performed in CO<sub>2</sub>RR [46]. As a comparison, the intensity of positive characteristic peak representing the accumulated water molecules at 1670  $\text{cm}^{-1}$  in control NCA was much higher than that in P@NCA (Fig. 4d-e), indicating the significantly

accelerated water dissociation process in P@NCA with introducing P species. Notably, the \*CO intermediates (1960  $\text{cm}^{-1}$ ) adsorbed on P@NCA aerogel and NCA are continuously consumed and accumulated, respectively, suggesting the P@NCA aerogel possess an accelerated \*CO desorption step compared with NCA in CO<sub>2</sub>RR. To further confirm the accelerating effect of P@NCA aerogel on water dissociation with the incorporation of P, the kinetic isotope effect (KIE) were investigated. In Fig. 4f, the KIE value of P@NCA aerogel was calculated to be 1.1, which is lower than that of NCA (1.5), revealing the processes of water dissociation and subsequent proton transfer are promoted with the assistance of P species [47]. Meanwhile, according to the Arrhenius equation, the activation energies of P@NCA aerogel and NCA are calculated to be 10.33 and 32.50  $\text{kJ mol}^{-1}$ , respectively (Fig. 4g and Fig. S32) [48,49], indicating a reduced energy required for CO<sub>2</sub>-CO conversion under the synergistic effect of N and P species. Based on the above analysis, the corresponding CO<sub>2</sub>RR mechanism is proposed (Fig. 4h and Fig. S33): under the assistance of P atoms, the dissociation of water molecules is obviously accelerated on P@NCA aerogel, which results in a rapid protonation process of CO<sub>2</sub> to \*COOH. Subsequently, in the final CO desorption step, the doped P atoms promotes the more quickly desorption of \*CO intermediate from real active sites of pyrrolic-N, thus boosting the whole CO<sub>2</sub>RR activity of P@NCA aerogel.

In order to reveal the intrinsic mechanism of P@NCA aerogel in CO<sub>2</sub>RR, three different types of N, P co-doped graphene constructions were studied by first principles calculation, including graphitic N-P, pyridinic N-P and pyrrolic N-P (Fig. 5a), and the calculation details were given in Supporting Information. Due to the radius of P atom is larger than N and C atoms, the plane of graphene is wrinkled with P atom protruding out. To clarify the reaction thermodynamic behavior, the CO<sub>2</sub>-to-CO conversion is evaluated by calculating the Gibbs free energies as following elementary steps



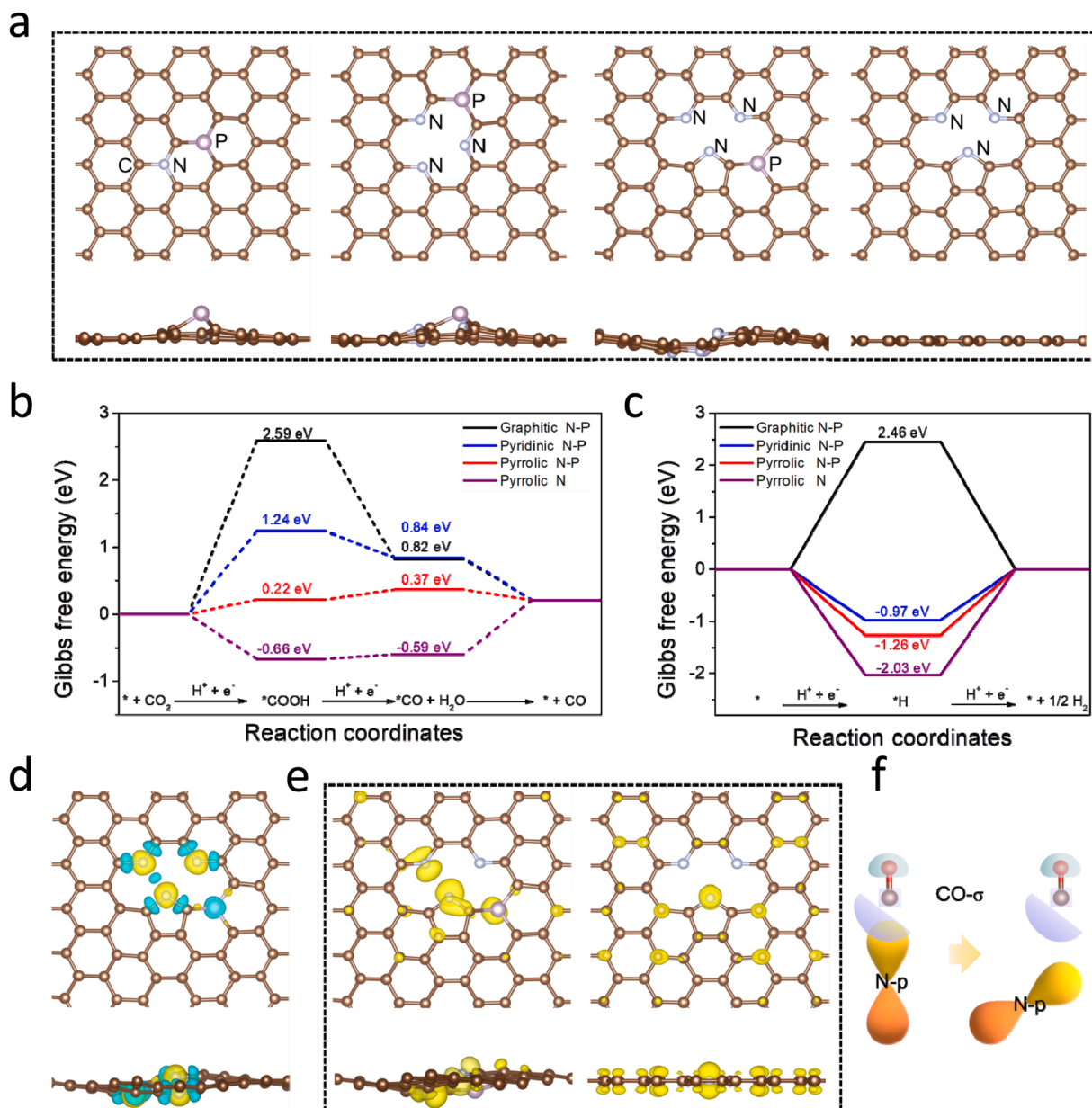


**Fig. 4.** (a–b) In situ ATR-FTIR spectra and corresponding contour image of P@NCA tested from  $-0.3$  V to  $-1.2$  V, (c)  $J_{\text{CO}}$  of P@NCA at different  $\text{KHCO}_3$  concentrations, (d–e) in situ ATR-FTIR spectra and corresponding contour image of NCA tested from  $-0.3$  V to  $-1.2$  V, (f)  $J_{\text{CO}}$  and KIE values of H/D for NCA and P@NCA, (g) the activation energies for P@NCA and NCA in  $\text{CO}_2\text{RR}$ , (h) mechanism diagram of  $\text{CO}_2\text{RR}$  for P@NCA and NCA.

As shown in Fig. 5b, the maximum reaction energy steps are the formation of \*COOH intermediate for graphene doped with graphitic N-P (2.59 eV), pyridinic N-P (1.24 eV), and pyrrolic N-P (0.22 eV). As a result, the pyrrolic-N of graphene doped with pyrrolic N-P presents an excellent activity for  $\text{CO}_2\text{RR}$ . It is noting that, the pyrrolic-N without P doping exhibits the much stronger adsorption for \*COOH and \*CO intermediates than the pyrrolic N-P, which induces to the large reaction energy ( $\Delta G = 0.81$  eV) of CO desorption step for NCA. In addition, the thermodynamics of competitive HER is evaluated in Fig. 5c. The calculation results show that the \*H adsorption strength on pyrrolic-N of graphene is too strong, which leads to the inhibition of the adsorption of \*COOH and \*CO intermediates. With P doping nearby, the strength of \*H adsorption on pyrrolic-N is decreased markedly, contributing the formation of \*COOH intermediate and CO selectivity of  $\text{CO}_2\text{RR}$ . To uncover the effect of P doping in reactivity, charge density difference is further analyzed between pyrrolic-N with and without P surrounding. It is clearly seen that the doping of P atom efficiently induces the increase of charge density for pyrrolic-N (Fig. 5d). Moreover, the electronic structure of pyrrolic-N is modified by P doping shown in Fig. 5e. The orbital wave function (squared) close to the Fermi level (Fig. 5f) shows that the pyrrolic-N site of NCA has dominantly occupied orbital contribution with perpendicular orientation to the layer, which is favorable to strong electronic coupling with maximum orbital overlap and consequently strong intermediates adsorption [50]. With P doping, the orientation of orbital wave function close to the Fermi level incline,

inducing the reduction of components outside the vertical plane. As a result, the hybridization between the molecular orbital of \*CO intermediate and the p orbital of N atom is weakened. Therefore, the decreased adsorption strength for CO on pyrrolic N-P favors the last elementary reaction step, i.e., \*CO desorption.

Inspired by the outstanding  $\text{CO}_2\text{RR}$  performance of P@NCA aerogel, an aqueous Zn- $\text{CO}_2$  battery with P@NCA aerogel cathode and Zn plate anode was assembled to evaluate the practical application ability of  $\text{CO}_2\text{RR}$  (Fig. 6a). The 6.0 M KOH containing 0.2 M  $\text{Zn}(\text{CH}_3\text{COO})_2$  and  $\text{CO}_2$ -saturated 0.5 M  $\text{KHCO}_3$  solutions were used as anolyte and catholyte, respectively [51]. In Fig. 6b, the discharge voltage curve showed an open-circuit voltage of 0.8 V and a high power density of  $0.8 \text{ mW cm}^{-2}$  for P@NCA aerogel. In addition, the P@NCA aerogel presented a maximum CO FE of 92 % at the discharge current density of  $2.0 \text{ mA cm}^{-2}$  (Fig. 6c), demonstrating the superior  $\text{CO}_2\text{RR}$  activity in Zn- $\text{CO}_2$  battery. Based on the good power density of P@NCA aerogel, two Zn- $\text{CO}_2$  batteries connecting in series can light up a red light-emitting diode in Fig. 6d, suggesting a large potential of P@NCA aerogel for energy conversion. Meanwhile, the galvanostatic charge-discharge cycling test of P@NCA aerogel was further performed at  $0.5 \text{ mA cm}^{-2}$ , in which no obviously enlarged voltage gap was observed within 20 h (Fig. 6e), demonstrating the excellent durability.



**Fig. 5.** (a) The constructions of graphene with graphitic N-P, pyridinic N-P, pyrrolic N-P and pyrrolic N. The Gibbs free energy profiles of (b) CO<sub>2</sub>-CO and (c) HER reaction. (d) The charge density differences between pyrrolic N-P and pyrrolic N, iso-value 0.03 |e| Å<sup>-3</sup>. (e) The occupied orbital distribution near Fermi-level for graphene doped with pyrrolic N-P and pyrrolic N. (f) The schematic diagram for orbital interaction with different orientation.

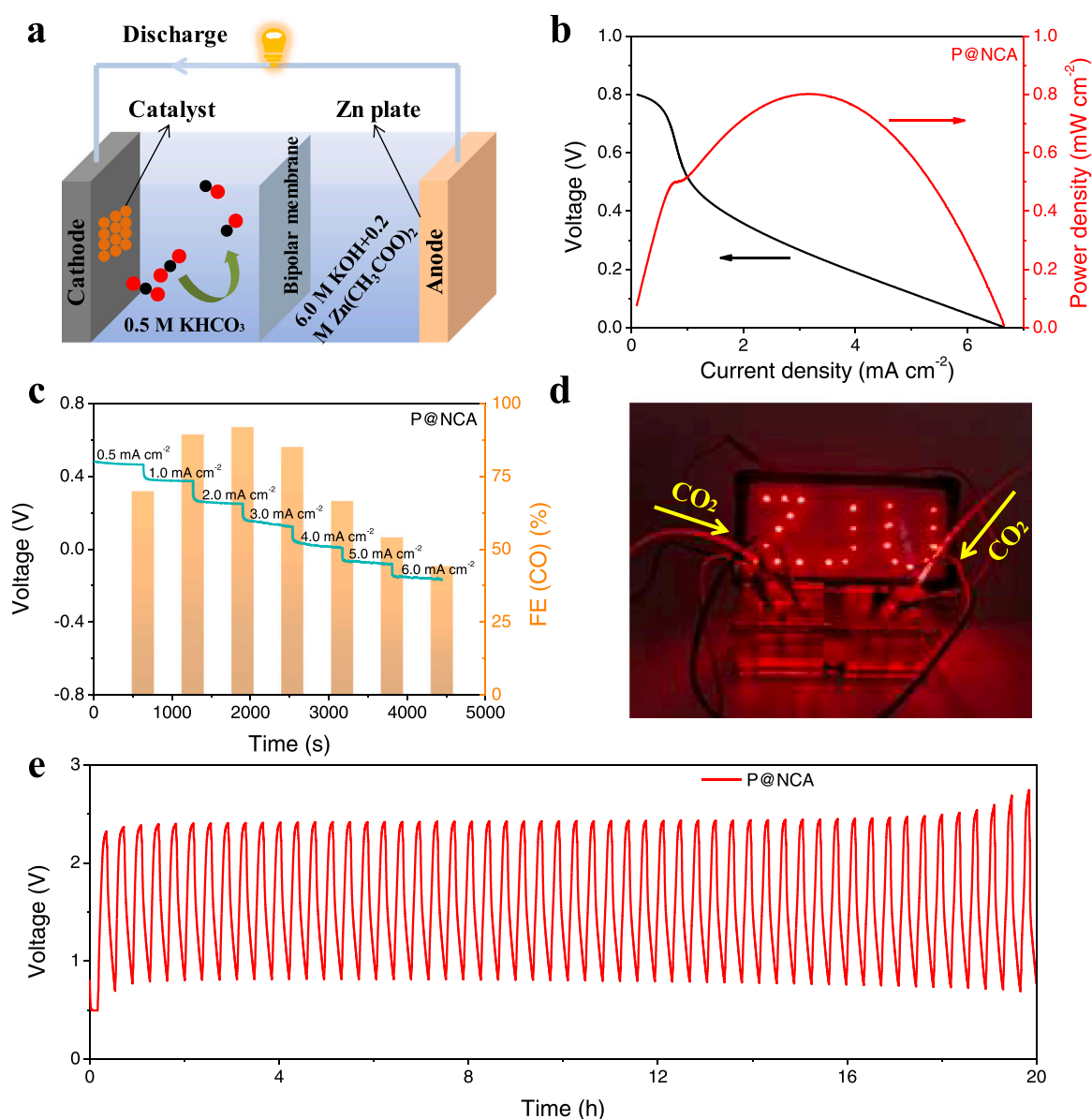
#### 4. Conclusions

In summary, we have developed a new type of metal-free P@NCA aerogel catalyst with uniformly dispersed P, N sites and 3D mesh-like structure, that demonstrates highly efficient CO<sub>2</sub>RR performance. A wide potential window from -0.3 V to -0.9 V with CO FE of above 90 % was achieved on P@NCA aerogel, which exceeded the most of the previously reported metal-free carbon based CO<sub>2</sub>RR catalysts. More importantly, the P@NCA aerogel delivered a superb CO<sub>2</sub>RR activity with a high CO FE of 85 % under 100 mA cm<sup>-2</sup>. Experimental observations and in situ ATR-FTIR results confirmed the real active site of pyrrolic-N species on P@NCA aerogel, while the P atoms anchored around the pyrrolic-N site played a dominant role in accelerating water dissociation to boost the proton transfer rate, thus promoting the formation of key \*COOH intermediate. Further theoretical simulation results are in good agreement with our experiments, which allows us to identify the reaction mechanism for how P doping affects CO<sub>2</sub>RR activity and selectivity.

Furthermore, a rechargeable Zn-CO<sub>2</sub> battery with P@NCA aerogel cathode realized a maximum power density of 0.8 mW cm<sup>-2</sup>. The accelerating proton feeding via P doping method reported in this work opens up a horizon for designing the highly efficient metal-free carbon based catalyst useful for CO<sub>2</sub>RR and other electrochemical reactions, including N<sub>2</sub> reduction, O<sub>2</sub> reduction, and water electrolysis.

#### CRediT authorship contribution statement

**Wanzhen Zheng:** Conceptualization, Methodology, Investigation, Writing – original draft, Writing – review & editing. **Dashuai Wang:** Conceptualization, Investigation, Software. **YiKai Zhang:** Methodology, Investigation, Writing – review & editing. **Sixing Zheng:** Investigation. **Bin Yang:** Investigation. **Zhongjian Li:** Investigation. **Raul D. Rodriguez:** Investigation. **Tao Zhang:** Investigation. **Lecheng Lei:** Investigation, Funding acquisition. **Siyu Yao:** Supervision. **Yang Hou:** Conceptualization, Writing – review & editing, Resources, Supervision,



**Fig. 6.** (a) Schematic configuration of Zn-CO<sub>2</sub> battery, (b) discharge curve and power density of P@NCA, (c) CO FE and current density of P@NCA in Zn-CO<sub>2</sub> battery, (d) digital photo of light-driven by two Zn-CO<sub>2</sub> batteries, (e) galvanostatic charge-discharge cycling curve of P@NCA at 0.5 mA cm<sup>-2</sup>.

Funding acquisition.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data Availability

Data will be made available on request.

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#### Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.nanoen.2022.107980](https://doi.org/10.1016/j.nanoen.2022.107980).

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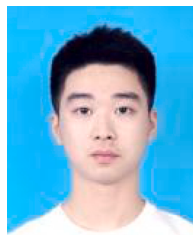
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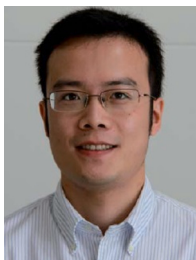
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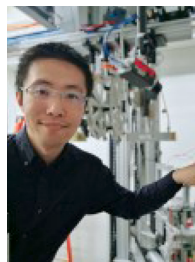
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