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Constructing BiOBr/TiO₂ heterostructure nanotubes for enhanced adsorption/photocatalytic performance

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ABSTRACT

The fabrication of heterostructure materials with the synergy of adsorption and photocatalysis is an effective strategy to promote the removal of organic pollutants. In this work, BiOBr/TiO₂ nanotubes were successfully obtained to enhance the removal of organic pollutants from water via the synergy of adsorption and photocatalysis. The TiO₂ nanotubes were prepared using waste foam as the primary raw material by impregnation calcination, and BiOBr nanosheets were grown on their surface by a simple solvothermal method. The BiOBr/TiO₂ nanotubes have significantly enhanced separation efficiency of photo-generated carriers, synergistic adsorption, and photocatalytic decomposition compared with pure TiO₂ nanotubes. The enhanced activity is mainly attributed to the growth of BiOBr nanosheets leading to a substantial increase of the specific surface area and the internal electric field between the heterojunction of BiOBr/TiO₂. In addition, possible degradation pathways for organic pollutants are proposed based on trapping experiments, electron spin resonance spectrometer, and density functional theory calculations. Our work provides a promising strategy for combining the advantages of adsorption and photocatalytic technologies for environmental remediation, and we anticipate it will be of significant interest to researchers in environmental science, materials science, and chemical engineering.

1. Introduction

In recent years, antibiotics have been crucial in preserving human health. However, their excessive use also has several negative effects, including upsetting the ecological balance, endangering human health, and gravely disrupting the ecosystem [1–4]. Tetracycline hydrochloride (TCH), which is made from tetracycline and hydrochloric acid, has a noticeably increased water solubility [5]. As a refractory antibiotic agent, it might be stable in a water environment for an extended period of time [6-9]. However, conventional wastewater treatment methods, such as adsorption, precipitation, and biological treatment are difficult to achieve satisfactory processing efficiency, and the released effluent still threatens aquatic life and human health [10-13]. Generally speaking, adsorption is a cost-effective method of treating sewage [14,15]. However, the adsorption process does not eliminate environmental pollutants, which merely moves them from one phase to another, frequently resulting in secondary contamination [16,17]. Photocatalysis is regarded as a promising advanced oxidation processes (AOPs) for the oxidation or even mineralization of organic pollutants due to its high efficiency, low secondary pollution, ease of use, and low cost of treatment [18–20]. Titanium dioxide (TiO₂) has received much long-term attention in the process of degrading water contaminants [21–24]. However, due to its high rate of photogenerated charge recombination, it is difficult to employ sunlight to effectively destroy contaminants in industry [25–27]. In addition, the TiO₂ photocatalyst can only be stimulated by ultraviolet (UV) light because of the broad band gap energy (Eg = 3.2 eV), which accounts for only about 5 % of the solar energy [28–30]. Therefore, developing innovative techniques to produce TiO₂ with high solar usage and charge separation efficiency is crucial [31–33].

Bismuthyl bromide (BiOBr) is a type of layered oxide semiconductor containing tetragonal ${\rm [Bi_2O_3]^{2+}}$ slabs sandwiched within double Br $^-$ layers, which has been extensively researched for its excellent chemical stability, narrow band gap (Eg = 2.69 eV), and excellent photocatalytic activity [34,35]. Recent research works investigated the construction of the heterojunction structure of BiOBr and TiO2 to enhance

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photocatalytic performance, which could increase the visible light utilization efficiency and inhibit the recombination of photogenerated charge carriers [36,37]. For example, Yang et al. controlled the growth of BiOBr on TiO2 nanorods, resulting that BiOBr/TiO2 heterostructure displayed an enhanced photocatalytic performance in the decomposition of RhB under visible light [38]. Zhou et al. prepared BiOBr/TiO2 heterojunction photocatalysts, which demonstrated excellent visiblelight photocatalysis with 1.34 times and 12.63 times higher than that of bulk BiOBr and TiO2, respectively [39]. However, developing lowcost, simple processes, and high-yield BiOBr/TiO2 photocatalysts with higher photocatalytic activity is still challenging [40-42]. Compared to bulk semiconductor materials, controlling the nanoscale morphology has been employed to optimize the performance of photocatalysts [34,43]. Additionally, combining the adsorption and photocatalytic performance of the semiconductor heterostructure is also an effective approach for efficiently removing pollutants [44–46].

Herein, we constructed an adsorption-photocatalysis dual-functional photocatalyst by the ${\rm BiOBr/TiO_2}$ heterojunction material. The uniform growth of BiOBr nanosheets on ${\rm TiO_2}$ nanotubes not only maintains the morphology stability, but also reduce the recombination of charge carriers by the heterostructure. The TCH was used as a model pollutant to research the adsorption and catalytic performances of the ${\rm BiOBr/TiO_2}$ nanotubes. Furthermore, the degradation mechanism of TCH was elucidated by free radical trapping experiments, electron spin resonance spectrometer (ESR), and density functional theory (DFT) calculations. This work will provide a simple and efficient method for manufacturing hollow heterojunction nanotubes, designing dual-functional adsorbent/photocatalysts and developing an excellent potential application for solving environmental problems.

2. Experimental section

2.1. Materials

Waste foam (the main component is polystyrene, PS) was collected from the outer package of Aladdin Industrial Corporation. Tetrabutyl titanate (TBOT, 97 %), bismuth nitrate pentahydrate (Bi(NO₃)₃·5H₂O, 99 %), hexadecyl trimethyl ammonium bromide (CTAB, 99 %), tetracycline hydrochloride (TCH, 99 %), methyl orange (MO, 96 %), rhodamine B (RhB, 99 %), 1,4-benzoquinone (BQ, 99 %), disodium ethylenediaminetetraacetate (Na₂EDTA, 99 %), and tert-butyl alcohol (t-BuOH, 99 %) were purchased from Aladdin Industrial Corporation. *N*, *N*-dimethylformamide (DMF, 99 %), ethylene glycol (EG, 99 %), methylene blue (MB, 97 %) and absolute ethanol were obtained from Beijing chemical works. All chemicals used in experiments were of analytical grade and used without further purification.

2.2. Preparation of TiO2 nanotubes

 TiO_2 nanotubes were prepared by electrospinning combined with improved impregnation calcination. Firstly, 1.25 g PS was dissolved in 5 mL DMF to obtain the precursor solution. The PS nanofibers were manufactured using electrospinning equipment with a high voltage (16 kV) power supply. The ambient temperature was 25 °C and the ambient humidity was 40 %. The distance from the injector and acceptor was 14 cm, and the flow rate of the precursor solution was controlled at 0.5 mL/h to obtain the PS nanofibers membrane. Then the collected membrane was soaked in TBOT/ethanol (including 1 mL TBOT and 10 mL ethanol) solution for 1 min, and dried in an air oven at 60 °C for 2 h. Finally, the TBOT/PS nanofibers membrane was calcined at a heating rate of 2 °C/min to 550 °C and kept for 2 h, and let cool at room conditions after annealing to obtain TiO_2 nanotubes.

2.3. Preparation of BiOBr/TiO2 heterostructures

0.4 g Bi(NO₃)₃·5H₂O and 0.45 g CTAB were firstly dissolved in 80 mL

EG ethanol solution. Then, the above solution and a certain amount of ${\rm TiO_2}$ nanotubes were transferred to a 100 mL Teflon-lined stainless-steel autoclave, sealed and heated for 4 h at 160 °C. Finally, the products were separated by centrifugation and washed with ethanol and deionized water three times (Scheme 1). The samples prepared with ${\rm TiO_2}$ content of 150 mg, 100 mg, and 50 mg were named TB1, TB2, and TB3, respectively. As the control sample, pure BiOBr was synthesized with the same procedures without adding ${\rm TiO_2}$ nanotubes.

The characterization instruments and parameters for the as-prepared photocatalysts are listed in the supplementary information.

2.4. Evaluation of adsorption activity

TCH has been widely used as a common antibiotic with low biodegradability in water. In this work, TCH (20 mg/L) was used as a simulated organic pollutant to evaluate the adsorption properties of the BiOBr/TiO $_2$ nanotubes. 40 mL TCH was put into a quartz tube, and 40 mg as-prepared catalysts were added into each quartz tube. The suspension was magnetically stirred in the dark, and 3 mL was extracted for centrifugation to measure the absorbance.

The detailed testing procedures for adsorption properties are presented in the supplementary information.

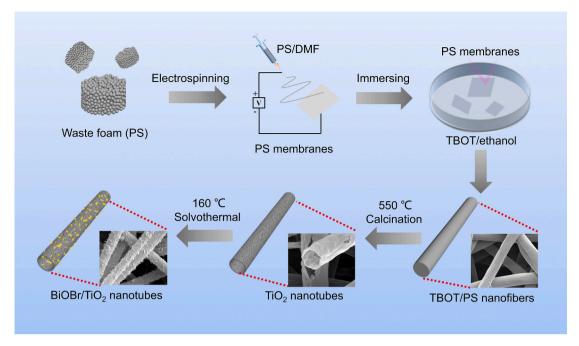
2.5. Evaluation of photocatalytic activity

The photocatalytic process of the as-prepared samples was evaluated by degradation of TCH. Firstly, 40 mg samples were mixed with 40 mL TCH (20 mg/L) in a quartz tube and stirred in the dark for 4 h to achieve the adsorption/desorption equilibrium. Then, the mixed dispersion was irradiated under a xenon light source (300 W); the distance between the tube and the xenon light was 10 cm. The content of TCH could be measured by UV–vis spectroscopy (Shimadzu, UV-2550) after being collected from the mixture (3 mL) and centrifuged at regular time intervals.

3. Results and discussion

Fig. 1a shows the scanning electron microscopy (SEM) of PS fibers with a diameter of 600-900 nm. After soaking and heating treatment, the hollow TiO2 nanotubes with a diameter of about 800 nm were successfully fabricated, as shown in Fig. 1b-c. Compared with the pure TiO₂ nanotubes, the BiOBr/TiO2 samples retain the nanotube morphology. Based on the observations from Fig. 1d-f, as the concentration of TiO₂ nanotubes decreases in the solvothermal reaction, there is a corresponding increase in the density of BiOBr nanosheets that cover the surface of the TiO₂ nanotubes. In Fig. 1e, the TiO₂ nanotubes marked as TB2 exhibit the most uniform and well-dispersed BiOBr nanosheets. This indicates that the quantity of the base material plays a crucial role in determining the morphology of the BiOBr nanosheets on the TiO2 nanotubes. Fig. 1g shows the transmission electron microscopy (TEM) image of the TB2 nanotube, and a typical nanosheet structure of the BiOBr could be observed. In the HRTEM image of TB2 (Fig. 1h), the interplanar lattice spacing of 0.284 nm and 0.353 nm are found in the composite nanotubes, which corresponds to the (102) plane of BiOBr (JCPDS No. 09-0393) and the (101) plane of anatase TiO₂ (JCPDS No. 21-1272), respectively. It also confirms that BiOBr nanosheets grow on the TiO₂ nanotubes through a simple solvothermal reaction to form the heterostructure. Meanwhile, the energy dispersive X-ray (EDX) elemental mapping (Fig. 1i) corresponding to Fig. 1e further demonstrates that the BiOBr nanosheets are uniformly dispersed on the surface of TiO2 nanotubes.

The X-ray diffraction (XRD) patterns of the BiOBr, TiO₂, and BiOBr/ TiO₂ (TB2) nanotubes are shown in Fig. 2a. The diffraction peaks located at 25.34° , 37.83° , 48.13° , 55.13° , and 62.67° correspond to the (101), (004), (200), (211), and (204) lattice planes of the TiO₂ anatase phase (JCPDS No. 21–1272). Pure BiOBr exhibits peaks at 31.62° and 32.21° ,



Scheme 1. Schematic for the fabrication of TiO₂ and BiOBr/TiO₂ nanotubes.

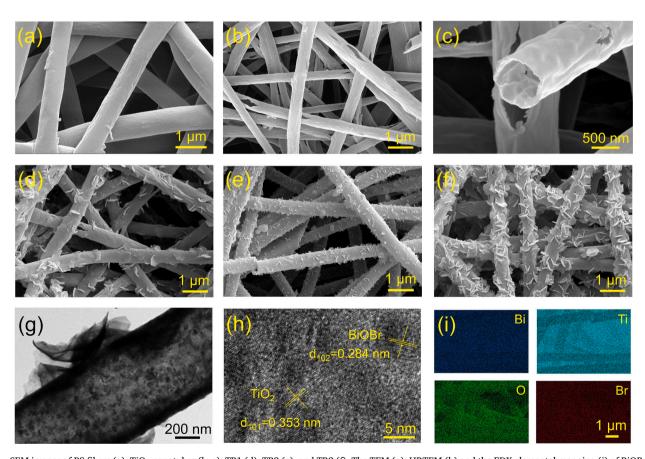


Fig. 1. SEM images of PS fibers (a), TiO_2 nanotubes (b, c), TB1 (d), TB2 (e), and TB3 (f). The TEM (g), HRTEM (h) and the EDX elemental mapping (i) of $BiOBr/TiO_2$ (TB2) nanotubes.

which can be attributed to the (102) and (110) lattice planes of the standard tetragonal phases BiOBr (JCPDS No. 09–0393). The BiOBr/TiO $_2$ nanotubes show the peaks of TiO $_2$ anatase and the BiOBr tetragonal phases. As shown in Fig. S1, the BiOBr diffraction peaks intensity

strengthens as the relative concentration of Bi ions in the reaction increases. The sharp and intense diffraction peaks indicate that the obtained photocatalysts have good crystallinity and high purity.

Fig. 2b presents the X-ray photoelectron spectroscopy (XPS) results

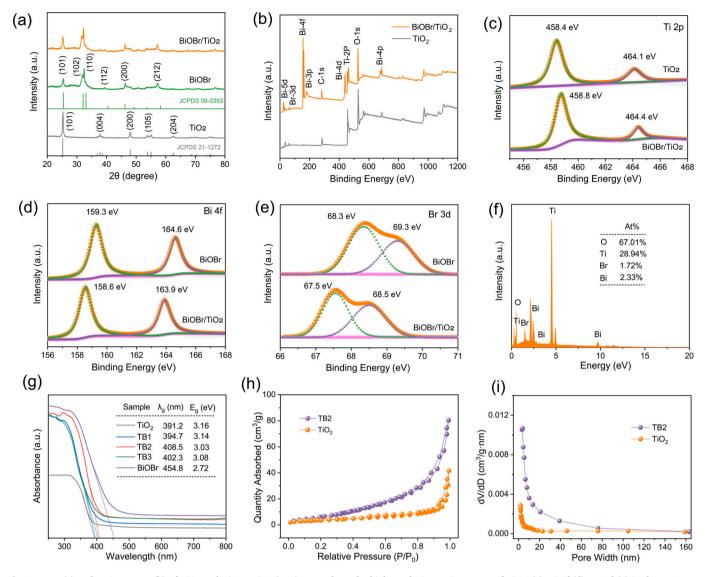


Fig. 2. XRD (a) and XPS patterns (b) of TiO_2 and $BiOBr/TiO_2$ (TB2) nanotubes. The high-resolution XPS patterns of Ti 2p (c), Bi 4f (d), Br 3d (e) in the TB2. EDX spectrum (f) of TB2 nanotubes. UV–vis diffuse reflectance spectra (g) and the deduced band gap energy of the samples. N_2 adsorption-desorption isotherms (h) and pore size distribution plots (i) of the TiO_2 and TiO_3 .

of TiO_2 and $BiOBr/TiO_2$ (TB2) nanotubes. The survey XPS spectrum of TB2 clearly displays the peaks corresponding to Ti, O, Bi, and Br, confirming the presence of BiOBr and TiO2. Moreover, the high-resolution XPS spectrum of Ti 2p for TB2 (Fig. 2c) reveals two peaks located at binding energies of 458.8 eV and 464.4 eV, corresponding to Ti 2p_{1/2} and Ti 2p_{3/2}, respectively. The distance between the two peaks is 5.6 eV, demonstrating that Ti⁴⁺ is the main chemical state in TiO₂. Significantly, the binding energies of Ti for BiOBr/TiO2 (TB2) nanotubes are 0.3-0.4 eV higher than that of TiO2, which means that the electrons are transferred from TiO2 to BiOBr when the BiOBr/TiO2 heterojunction is formed. The high resolution XPS of Bi 4f (Fig. 2d) for TB2 shows two peaks at 158.6 and 163.9 eV. These peaks correspond to Bi $4f_{7/2}$ and Bi $4f_{5/2}$, which are the characteristic peaks of Bi^{3+} in BiOBr. The Br 3d spectrum (Fig. 2e) shows two peaks at 67.5 and 68.5 eV, corresponding to the Br 3d_{5/2} and Br 3d_{3/2} in TB2, respectively. The EDX spectrum of TC2 (Fig. 2f) demonstrates that Bi, O, Br, and Ti elements all exist in the composite nanotubes, and the corresponding atomic percentage are 2.33 %, 67.01 %, 1.72 % and 28.94 %, respectively. These collective XPS and EDS results indicate that the composite photocatalyst combines BiOBr and TiO2 with an electron transfer process between these two components.

The UV–vis diffuse reflectance spectra (DRS) of the TiO_2 , $BiOBr/TiO_2$, and BiOBr were recorded to analyze their light absorption properties. As shown in Fig. 2g, the pure TiO_2 nanotubes can only absorbs the UV light, the BiOBr and $BiOBr/TiO_2$ nanotubes can absorb UV light and an amount of visible light. The absorption intensity of $BiOBr/TiO_2$ nanotubes in the whole visible light range is significantly enhanced, which is due to the heterostructure between the BiOBr and TiO_2 . The E_g values of the nanotubes are shown in the inset of Fig. 2g by the function [25]:

$$E_g=1240/\lambda_g$$

The E_g values of TiO₂, TB1, TB2, TB3, and BiOBr were estimated to be 3.16, 3.14, 3.03, 3.08, and 2.72 eV, respectively. Clearly, the band gaps of BiOBr/TiO₂ nanotubes are smaller than that of pure TiO₂, which should be associated with BiOBr/TiO₂ heterostructure formation. Therefore, the BiOBr/TiO₂ nanotubes have an improved photocatalytic performance due to better visible light absorption.

Generally speaking, larger specific surface areas and pore volumes could provide more favorable conditions for photocatalytic processes. Therefore, the N_2 absorption-desorption isotherms of the as-prepared samples were obtained. As shown in Fig. 2h, the isotherms of TiO_2

and TB2 nanotubes all belong to type IV isotherm, indicating the existence of mesopores in the nanotubes. Meanwhile, the hysteresis loop of the TiO $_2$ and TB2 all belong to type H3, which implies the existence of narrow slit-shaped mesopores. Compared with pristine TiO $_2$ nanotubes, the TB2 nanotubes have a larger Brunauer-Emmett-Teller (BET) specific surface area from 14.3 $\rm m^2/g$ to 26.5 $\rm m^2/g$ as shown in Fig. 2h. The BiOBr/TiO $_2$ nanotubes possess an increased specific surface area, translating to more available surface reaction sites for the adsorption and decomposition of pollutant molecules. This characteristic ultimately leads to an improvement in photocatalytic activity.

The photocatalytic performance of the TiO2 and BiOBr/TiO2 nanotubes was evaluated by the degradation of TCH under xenon light irradiation, and the photocatalytic degradation efficiency of TCH was calculated by C_t/C₀ (C_t and C₀ correspond to the concentration of TCH at the time t, and the initial concentration of TCH, which conversion from the UV-vis spectroscopy by the Lambert-Beer law). In the dark, all samples reached the adsorption/equilibrium after two hours, and the pure TiO₂ nanotubes exhibited the lowest removal rate by adsorption (Fig. 3a). The adsorption rates (Fig. 3b) of TiO₂, TB1, TB2, and TB3 were estimated to be 24.4 %, 47.8 %, 55.7 %, and 64.1 %, which showed an upward trend with the increasing of BiOBr loading. After the adsorption/equilibrium under dark conditions, all samples were exposure to xenon light for 180 min. With the mass ratio of BiOBr increases, the photocatalytic activity of the as-prepared samples in the degradation of TCH first increased and then decreased (Fig. 3c), the TB2 reaching a maximum 88.1 %, which is higher than the pure TiO₂ (40.9 %), TB1 (76.9 %), and TB3 (82.3 %). The concentration of TCH does not change significantly without photocatalysts. In addition, the pseudo-first-order kinetic analysis is applied to compare the reaction kinetics of the degradation rate of TCH as follows [7]:

$$\mathit{ln}(C_0/C_t) = kt$$

where k is the pseudo-first-order rate constant, which could be used to measure the removal rate. As shown in Fig. 3d, the k values follow the order of TB2 $(0.0074~\text{min}^{-1}) > \text{TB3}~(0.0039~\text{min}^{-1}) > \text{TB1}~(0.0047~\text{min}^{-1}) > \text{TB2}~(0.0047~\text{min}^{-1}) > \text{TB2}~(0.0047~\text{min}^{-1}) > \text{TB2$

 $\rm min^{-1}) > \rm TiO_2$ (0.00128 min $^{-1}$). The TC2 nanotubes exhibit the highest photocatalytic activity, which is 5.8 times higher than that of pure TiO_2 nanotubes. The improved photocatalytic efficiency of BiOBr/TiO_2 nanotubes may be due to the following reasons. First, the high specific surface area increases the active site of the reaction. Second, the enhanced adsorption performance is more conducive to capturing pollutant molecules. Third, the heterostructure between the interface of BiOBr and TiO_2 provides more electron immigration pathways, promoting photogenerated electron-hole separation and transport.

The adsorption process has a significant effect on the removal of organic pollutants. Therefore, different types of adsorption experiments were performed under dark conditions to investigate the adsorption performance of BiOBr/TiO2 nanotubes, as shown in Fig. 4. The adsorption curves (Fig. 4a) of TCH by TB2 was estimated from the Fig. 3a, and the corresponding pseudo-first and pseudo-second-order models of the TB2 are shown in Fig. 4b and c. These results show that the pseudo-first-order dynamics description ($R^2 = 0.991$) is more consistent with the experimental process than the pseudo-second-order $(R^2 = 0.984)$. Therefore, the adsorption capacity of the composite nanotubes can be enhanced by an appropriate modification with BiOBr. To further explore the adsorption mechanism of BiOBr/TiO₂ nanotubes, the adsorption process of TCH in different initial concentrations by TB2 was studied (Fig. 4d), the data were analysed using the Langmuir (Fig. 4e) and the Freundlich (Fig. 4f) isotherm models. The result was better described by the Freundlich model based on its higher coefficient of determination ($R^2 = 0.992$), which indicated that the adsorption of TCH onto TB2 transpired via multilayer adsorption.

In general, the photocatalytic degradation process of the organic pollutants is mainly driven by different reactive oxygen species from the photocatalysts, such as holes (h^+), superoxide radical (\bullet O₂), and hydroxyl radical (\bullet OH). To better identify the active species driving the photocatalytic process, trapping experiments were performed for TCH degradation with TB2 nanotubes with Na₂EDTA, BQ, and t-BuOH used as scavengers for h^+ , \bullet O₂, and \bullet OH, respectively. Generally speaking, the specific scavengers can react with reactive oxygen species faster than

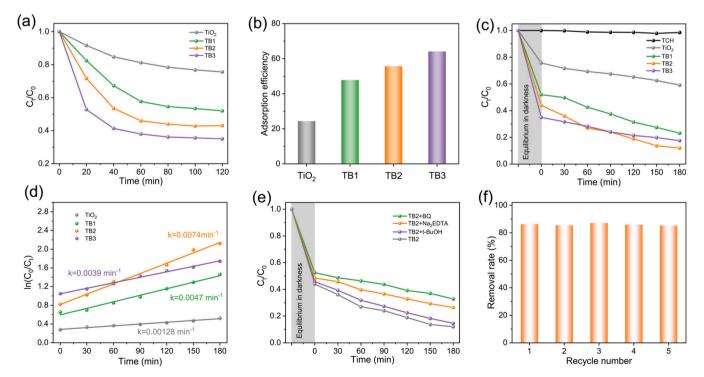


Fig. 3. The TCH adsorption curves (a) and adsorption rates (b) of the as-prepared samples in the darkness. The photocatalytic degradation of TCH (c) and the corresponding pseudo-first-order kinetic curves (d) of the as-prepared nanotubes under xenon light irradiation. The photocatalytic activity of TCH containing different scavengers (e) and the five recycling runs of TCH degradation over the TB2 (f).

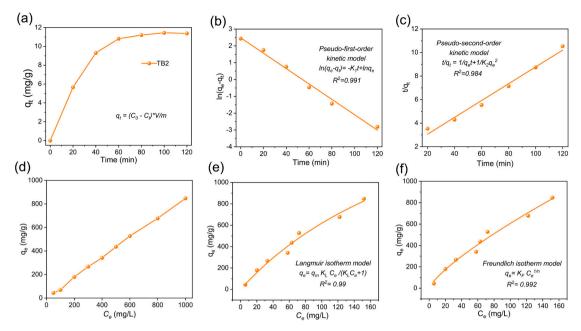


Fig. 4. The TCH adsorption curves (a), the pseudo first-order model (b) and pseudo second-order models for the TB2 (c). The TCH adsorption curves in different initial concentrations (d), the Langmuir (e) and Freundlich (f) isotherm models for the TB2.

organic pollutant molecules, resulting in the photodegradation rate decreased [5]. As shown in Fig. 3e, after adding BQ the photodegradation efficiency was significantly inhibited, which means the $\bullet O_2^-$ is the main active species in the process of TCH photodegradation. To further verify the active species from the trapping experiments, the ESR experiments were performed with 5,5-dimethyl-1-pyrroline N-oxide (DMPO) as the radical trapping agent. As shown in Fig. S2, no evident signal is observed in the dark, and show the obvious characteristic signal of $\bullet O_2^-$ under xenon lamp light condition, which means the $\bullet O_2^-$ is

reactive oxygen species in this system [47,48]. Meanwhile, the photodegradation efficiency of the TB2 was not significantly reduced after five cycles as shown in Fig. 3f, indicating that it has excellent stability during the photocatalytic reaction. Besides, the XRD pattern exhibit no significant change after the cycles as well (Fig. S3).

Common dyes MB, MO, and RhB were used as simulated contaminants to further analyze the adsorption and degradation properties of BiOBr/TiO₂ (TB2) nanotubes, and the removal rates (Fig. 5a) of the pollutants are about 67.9 %, 88.8 %, 92.8 % and 86.3 % corresponding

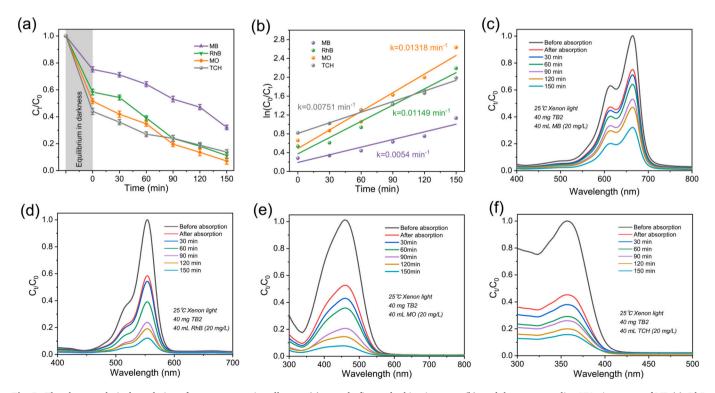


Fig. 5. The photocatalytic degradation of common organic pollutants (a), pseudo-first-order kinetic curves (b), and the corresponding UV–vis spectra of MB (c), RhB (d), MO (e) and TCH (f) over the TB2 nanotubes under xenon light irradiation.

to MB, RhB, MO and TCH by TB2, respectively. It is worth noting that the potential and relative molecular weight of the pollutants have great influence on the adsorption process. The organic pollutant molecule with negative potential (TCH, MO) showed stronger adsorption rate than that with positive potential (MB, RhB) by the TB2 nanotubes [20]. Based on our previous research, this phenomenon maybe caused by the [Bi₂O₂]²⁺ slabs in the BiOBr/TiO₂ nanotubes, which with a high positive potential and causes the potential adsorption [7]. Fig. 5b shows the pseudo-first-order kinetic analysis corresponding to the Fig. 5a, and the k values of the different pollutants shown a similar order with the adsorption rates, which means the adsorption process can directly affect the subsequent photodegradation, and the faster adsorption is advantageous to the degradation process. Fig. 5c-f shows the UV-vis spectra of MB, RhB, MO, and TCH by TB2 corresponding to the Fig. 5a. The absorption peak of the pollutants showed a significant decrease without red or blue shift, indicating that the pollutants were directly decomposed without intermediate products.

Generally, the photocurrent intensities directly reflect the ability of photogenerated charge transfer and separation. The photocurrent response curves of the samples under a xenon lamp with 50 s on-off cycles are presented in Fig. 6a. The photocurrent intensities display sharp rises and drops in sync with the light source turned on and off. The TB2 nanotubes exhibit a photocurrent density response obviously higher than that of pure TiO2 nanotubes, indicating that the BiOBr/TiO2 heterostructure nanotubes could efficiently promote the separation and transfer of photogenerated carriers under the xenon lamp light illumination. It is worth noting that the back part of photocurrent intensities TB2 go up increased obviously than the pure TiO2 nanotubes. This phenomenon may be caused by the unstable current of the xenon lamp source at startup, and the light response ability of the BiOBr/TiO2 nanotubes is stronger than the pure TiO2 nanotubes, which leads to significant fluctuations.

The photoluminescence (PL) emission spectra of TiO_2 and $BiOBr/TiO_2$ nanotubes are exhibited in Fig. 6b. Significantly, compared with pristine TiO_2 , the emission peak of $BiOBr/TiO_2$ nanotubes is

significantly weaker. Because the PL spectra could be related to the separation efficiency of photogenerated electron/hole pairs, and a lower emission intensity means a higher separation efficiency. Therefore, the recombination rate of photogenerated electrons and holes in BiOBr/TiO2 nanotubes is lower than in pristine TiO2 nanotubes. These results suggest that BiOBr nanosheets formed on the interface of TiO2 nanotubes could effectively promote the separation of photogenerated carrier and improve photocatalytic activity. Electrochemical impedance spectroscopy (EIS) curves of TiO2 and TB2 nanotubes are shown in Fig. 6c. The radius of the TB2 nanotubes is much smaller than pure TiO2, which means the charge transfer resistance in the BiOBr/TiO2 nanotubes is lower than that of the pure TiO2, facilitating separation and transfer of photogenerated carriers [49]. This phenomenon can be attributed to the heterojunction between BiOBr and TiO2 that effectively hinders the recombination of photogenerated carriers.

For further exploring the photogenerated carrier transfer mechanism, the Mott-Schottky plots were measured to investigate the flat potential of TiO2 and BiOBr nanotubes, as shown in Fig. 6d and e. The Mott-Schottky curves of the pure TiO₂ showing a positive slope indicate that the sample is an n-type semiconductor. Therefore, the conduction band (CB) positions can be estimated by the flat band potential, which is ascertained to be -0.71 V (vs. Ag/AgCl, pH = 7) for TiO₂ nanotubes based on the x-intercepts of Mott-Schottky curves [49]. Thus, the CB of TiO_2 is located at -0.51 eV (vs. NHE, pH = 0). According to the UV-vis DRS analysis of TiO_2 (3.16 eV) and the formula $E_g = E_{VB}-E_{CB}$, the valence band (VB) position of TiO_2 is calculated to be 2.65 eV (vs. NHE, pH = 0). Similarly, the CB and VB potentials of pure BiOBr are calculated to be 0.08 and 2.8 eV, respectively. Meanwhile, first-principles calculations of differential charge density (DFT) of BiOBr/TiO2 heterojunction were carried out and the result is shown in Fig. 6f. The yellow and blue regions represent net electron accumulation and depletion, respectively. Distinctly, the electron accumulates in the BiOBr part and the holes accumulate in the interface between TiO2 and BiOBr, which indicates that the BiOBr/TiO2 belongs to heterojunction in agreement with Mott-Schottky and XPS analyses.

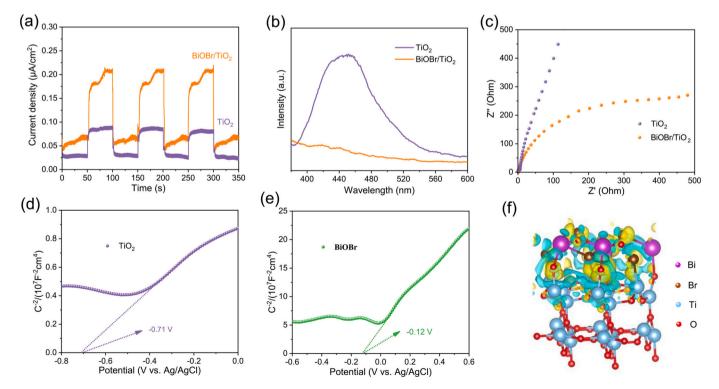


Fig. 6. The transient photocurrent response (a), PL spectra (b), and EIS (c) of TiO₂ and BiOBr/TiO₂ (TB2) nanotubes. The Mott-Schottky plots of TiO₂ (d) and BiOBr (e). The differential charge density (DFT) of BiOBr/TiO₂ (f).

Based on the results of trapping experiments, ESR, DFT calculations, and Mott-Schottky plots, a possible band energy diagram of BiOBr/TiO₂ Z-scheme heterojunction is displayed in Fig. 7, and a reasonable mechanism for the enhanced photoactivity of BiOBr/TiO2 is proposed. The growth of BiOBr on the surface of TiO2 leads to the transfer of electrons from TiO2 to BiOBr to achieve Fermi energy level equilibrium, while holes diffuse from BiOBr to TiO2 due to the difference in carrier density. This observation is consistent with the results obtained from DFT calculations [50]. Due to the charge redistribution between BiOBr and TiO2, an internal electric field from TiO2 to BiOBr is formed at the interface. Under xenon lamp irradiation, the photogenerated electrons in TiO2 and BiOBr are excited to their CB, and leave holes on the VB of TiO₂ and BiOBr simultaneously. The built-in electric field promotes the transfer of photogenerated electrons from BiOBr to TiO2 that combine with the photogenerated holes in TiO₂. Ultimately, the electrons in TiO₂ and the holes in BiOBr with strong redox ability remain spatially separated without recombination, contributing to enhanced photocatalytic performance.

4. Conclusion

In summary, $BiOBr/TiO_2$ nanotubes have been successfully synthesized by combining electrospinning, impregnation calcination and solvothermal method. Waste foam was used as raw material to save preparation costs. The $BiOBr/TiO_2$ nanotubes showed remarkable improvement in photo-generated carriers, synergistic adsorption, and photocatalytic decomposition for organic pollutants compared with pure TiO_2 nanotubes. The formation mechanism of $BiOBr/TiO_2$ nanotubes and the reason for enhanced photocatalytic performance were also elucidated. This research provides a new method for designing and preparing heterostructure nanotubes for inexpensive and efficient environmental remediation applications powered by sunlight.

Abbreviations

tetracycline hydrochloride TCH

electron spin resonance spectrometer ESR advanced oxidation processes AOPs titanium dioxide TiO2 bismuthyl bromide BiOBr ultraviolet UV electron spin resonance spectrometer ESR density functional theory DFT polystyrene PS tetrabutyl titanate TBOT bismuth nitrate pentahydrate Bi(NO₃)₃·5H₂O hexadecyl trimethyl ammonium bromide CTAB methyl orange MO rhodamine B RhB 1,4-benzoquinone BQ disodium ethylenediaminetetraacetate Na₂EDTA tert-butyl alcohol t-BuOH N, N-dimethylformamide DMF ethylene glycol EG methylene blue MB X-ray powder diffraction XRD scanning electron microscopy SEM energy dispersive X-ray analysis EDX transmission electron microscopy TEM X-ray photoelectron spectroscopy XPS diffuse reflectance spectra DRS Brunauer-Emmett-Teller BET holes \mathbf{h}^{+} superoxide radical •O₂ hydroxyl radical •OH 5,5-dimethyl-1-pyrroline N-oxide DMPO photoluminescence spectra PL electrochemical impedance spectroscopy EIS conduction band CB valence band VB

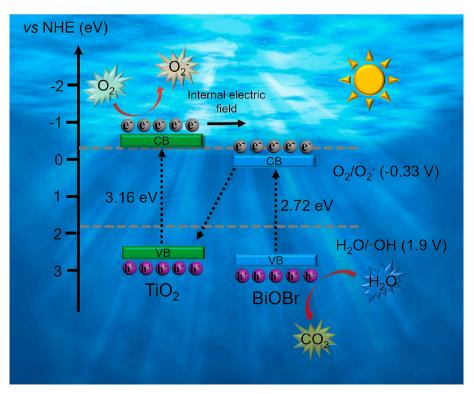


Fig. 7. Schematic representation of electron-hole separation and transportation in BiOBr/TiO2 nanotubes under simulated solar light irradiation.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

The data that has been used is confidential.

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Appendix A. Supplementary data

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